Vapor-Liquid Equilibria of Copolymer + Solvent and Homopolymer + Solvent Binaries: New Experimental Data and Their Correlation

Ram B. Gupta[†] and John M. Prausnitz*

Department of Chemical Engineering and Chemical Sciences Division, Lawrence Berkeley Laboratory, University of California, Berkeley, California 94720-1462

Sixty-four isothermal data sets for vapor-liquid equilibria (VLE) for polymer + solvent binaries have been obtained using a gravimetric sorption technique, in the range 23.5-80 °C. Solvents studied were acetone, acetonitrile, 1-butanol, 1,2-dichloroethane, chloroform, cyclohexane, hexane, methanol, octane, pentane, and toluene. Copolymers studied were poly(acrylonitrile-co-butadiene), poly(styrene-co-acrylonitrile), poly(styrene-co-butadiene), poly(styrene-co-butyl methacrylate), poly(vinyl acetate-co-ethylene), and poly(vinyl acetate-co-vinyl chloride). All copolymers were random copolymers. Some homopolymers were also studied: polyacrylonitrile, polybutadiene, poly(butyl methacrylate), poly(ethylene oxide), polystyrene, and poly(vinyl acetate). The composition of the copolymer may have a surprising effect on VLE. Normally, deviation from ideal behavior lies between those of the constituent homopolymers, according to the copolymer composition, as observed for cyclohexane + poly(ethylene-co-vinyl acetate) and chloroform + poly(styrene-co-butyl methacrylate). However, the strong nonideal behavior observed for systems containing hydrocarbons and poly(butadiene-co-acrylonitrile) shows that the effect of acrylonitrile is in excess of that expected from the copolymer composition. The perturbed hard-sphere chain (PHSC) equation of state was used to represent VLE of the copolymer solutions studied here.

Introduction

Vapor-liquid equilibria (VLE) in polymer + solvent systems are necessary for a variety of applications including surface acoustic-wave vapor sensors (Ballantine and Wohltjen, 1989; Grate et al., 1992), recovery of organic vapors from waste-air streams using a polymeric membrane (Baker et al., 1987; Matsumoto et al., 1991), pervaporation (Maeda et al., 1991) and other polymeric membrane-separation processes, polymer devolatilization (High and Danner, 1990), vapor-phase photografting (Kubota et al., 1983). For rational process and product design, we require experimental data and correlations based on molecular thermodynamics.

Copolymers are gaining an increasing amount of commercial interest because of their unique physical properties. World production of copolymers has been continuously rising over the last 20 years (Wohlfarth, 1993). However, published VLE studies for copolymer + solvent are rare. Some experimental results (Bonner and Prausnitz, 1974; Corneliussen et al., 1963; Iwai et al.,1990; Moore and Shuttleworth, 1963; Panayiotou and Vera, 1984) describe copolymer + solvent VLE for finite solvent concentrations. By contrast, there are about 1000 data sets published for homopolymer + solvent systems (Wohlfarth, 1994). There is a need for more experimental studies for copolymer solutions. This work contributes toward meeting that need.

The purpose of this work is to obtain VLE data for copolymer + solvent binaries and for some of their parent homopolymer + solvent binaries and to examine the effect of copolymer composition on VLE. The experimental data are correlated using an equation of state based on molecular thermodynamics.

Experimental Section

Apparatus and Procedure. Several experimental methods are available for measuring the vapor pressure

of a solvent in a polymer solution (Danner and High, 1993). These include inverse gas chromatography, piezoelectric sorption, differential vapor pressure, gravimetric sorption, light scattering, and turbidimetry analysis. We have used a gravimetric sorption technique because it gives direct measurement of the VLE of polymer + solvent systems. This classic method has been described in detail by Panayiotou and Vera (1984). Figure 1 gives a schematic of the apparatus. A copolymer or homopolymer sample in an aluminum pan is suspended by a sensitive, calibrated quartz spring in an evacuated glass chamber. The glass chamber is kept in a constant-temperature air bath. In a typical experiment, a known amount (25-50 mg) of polymer is loaded onto the pan and evacuated to remove any trace moisture, oligomers, or monomers present. Now the length of the spring is measured with a cathetometer to find the polymer mass, using a predetermined calibration curve. Solvent vapor is introduced into the evacuated system by quickly opening and closing the solvent valve. The system is allowed to equilibrate for several hours; subsequent pressure and spring-length readings yield pressure as a function of polymer-phase composition. Readings are taken with increasing pressure. All experiments are at pressures below 90% of the saturation vapor pressure to avoid solvent condensation in the system.

Quartz springs (Ruska Instruments Corp., Houston, TX) have a sensitivity of 1 mg/mm with the maximum load limit of 250 mg. These springs are equipped with reference pointers on the top and bottom of the spring. Quartz spring calibration (length versus mass, a linear correlation) is independent of temperature in the range of interest (23– 80 °C). The buoyancy effect on the calibration was also negligible. Spring length was measured using a cathetometer with a precision of 0.05 mm. Hence, mass was measured with a precision of 0.05 mg. The typical error in the measurements of mass fraction solvent in the solvent + polymer mixture, w_1 , was about 2%.

Pressure was measured using a mercury manometer kept at the same temperature as that of the polymer

^{*} To whom correspondence should be addressed.

^{*} Present address: Department of Chemical Engineering, Auburn University, Auburn, AL 36849.



Figure 1. Experimental apparatus for VLE measurement of polymer + solvent binary solutions. (Here VLE of two polymers with a common solvent can be measured simultaneously.)

solution. There was no solvent condensation on the mercury meniscus as the system pressure was always lower than the solvent saturation pressure. The pressure was measured with a precision of 0.1 kPa, except in the cases of cyclohexane (23.5 °C), octane (60 °C), and toluene (23.5 °C) where the precision was 0.01 kPa. The temperature of the air bath was controlled within 0.5 °C of the set point.

The reproducibility of the experimental procedure was checked by comparing our VLE data with literature values (Bawn et al., 1950; Kun et al., 1993) for the toluene + polystyrene system at 35 °C. Good agreement was observed.

Materials. All polymers were supplied by Scientific Polymer Products (SPP), Ontario, NY, except poly(butyl methacrylate) which was from Aldrich. These polymers are listed in Table 1 along with their molecular weight (M_w) , polydispersity (M_w/M_n) , and glass-transition temperature (t_g) . For copolymers, the mass percent compositions of the two constituent polymer segments are specified. All copolymers used here were random copolymers. Glasstransition temperatures for copolymers were calculated from the Fox-Flory equation

$$1/(t_{\rm g}+273.15) = F_1/(t_{\rm g1}+273.15) + F_2/(t_{\rm g2}+273.15) \tag{1}$$

where $t_g = \text{glass-transition temperature of the copolymer}$, t_{g1} and $t_{g2} = \text{glass-transition temperatures of parent}$ homopolymers, and F_1 and $F_2 = \text{mass fractions of mono-mers 1 and 2}$, respectively. All temperatures are in degrees Celsius.

Both copolymers and homopolymers had polydisperse molecular weights. These were used without further fractionation because the polymer molecular weight has negligible influence on the VLE behavior provided the molecular weight is above about 10 000 (see, for example, Kun et al. (1993) and Bonner and Prausnitz (1974)).

Solvents were used after degassing by a freeze/thaw procedure described by Panayiotou and Vera (1984).

Experimental Data. Table 1 gives a summary of all the new data obtained in this study. Table 2 gives experimental VLE data for polymer + solvent binary systems. Most of the experiments were performed at a temperature higher than the glass-transition temperature of the polymer. A total of sixty-four data sets were obtained

Table 1. Summary of New Experin	nental Vap	or-L	iquid	Equilibri	um Resu	lts for Poly	mer + So	lvent Systems							
	polym	er cha	racteri	stics											
polymer		1 M	t.d	catalog					temperatu	re/°C					
(wt % copolymer composition)	W,	Mn	ç	no. (SPP)ª	acetone	acetonitrile	1-butanol	1,2-dichloroethane	chloroform	cyclohexane	hexane	methanol	octane	pentane i	toluene
polyacrylonitrile	150 000		85	134		60		70, 80			60			60	
poly(21% acrylonitrile -co-butadiene)			-64	523		60				60	60		60	60	
poly(33% acrylonitrile -co-butadiene)			-50	528		60			60		60			60	
poly(51% acrylonitrile -co-butadiene)			-25	530		60				60	60		60	60	
poly(cis-1,4-butadiene)	$250\ 000$		-85	206		60			60	60	60			60	
poly(butyl methacrylate)			27						70						
poly(ethylene oxide)	$5\ 000\ 000$		-67	345					60						
polystyrene	$110\ 000$	1.1	100	580				70, 80							
poly(70% styrene-co-acrylonitrile)	$185\ 000$		95	495				70, 80							
poly(5% styrene-co-butadiene)			-80	199	50				50	60					35
poly(23% styrene-co-butadiene)			-60	200	50, 60				50	23.5, 60	70			60	35
poly(45% styrene-co-butadiene)			-30	201	50, 60				50	23.5, 60	70			60	
poly(50% styrene-co-butyl methacrylate)	$200\ 000$	1.9	56	595	.09				70			80			
poly(vinyl acetate)	194 000	3.1	30	024C			80		09						
poly(9% vinyl acetate-co-ethylene)		'	-118	506			80		09						
poly(25% vinyl acetate-co-ethylene)	88 000	4.4	-103	245					60	80					
poly(50% vinyl acetate-co-ethylene)	250 000	4.2	-74	785					60	80					
poly(70% vinyl acetate-co-ethylene)	280 000	5.6	-42	786					09	80					
poly(12% vinyl acetate-co-vinyl chloride)	30 000	2.1	75	063			80								
poly(10% vinyl acetate-co-vinyl chloride)	$115\ 000$		76	068								80			

Scientific Polymer Products, Ontario, NY

II

SPP

786 Journal of Chemical and Engineering Data, Vol. 40, No. 4, 1995

P/kPa P/kPa P/kPa P/kPa P/kPa P/kPa P/kPa P/kPa w_1 w_1 w_1 w_1 w_1 w_1 w_1 w_1 Acetone + Poly(styrene-co-butadiene) 1-Butanol + Poly(vinyl acetate) $t = 80 \ ^{\circ}\mathrm{C}$ Acetone + Poly(5% styrene-co-butadiene)2.70.005 0.036 0.092 8.0 12.315.20.012 $t = 50 \ ^{\circ}\text{C}$ 5.20.018 10.80.08113.90.011 13.70.003 45.30.067 51.10.078 59.6 0.115 26.50.023 46.5 0.07252.30.079 59.70.110 1-Butanol + Poly(vinyl acetate-co-ethylene) 36.3 0.040 47.20.067 53.3 0.080 61.3 0.120 41.50.056 47.30.066 55.20.091 1-Butanol + Poly(9% vinyl acetate-co-ethylene) Acetone + Poly(23% styrene-co-butadiene) $t = 80 \ ^{\circ}\text{C}$ 2.70.006 8.0 0.017 12.30.031 15.20.036 $t = 50 \ ^{\circ}\text{C}$ 5.20.01210.80.027 13.9 0.03446.5 26.50.024 0.075 52.30.088 59.6 0.12536.3 0.049 47.30.058 53.30.091 59.7 0.1201-Butanol + Poly(vinyl acetate-co-vinyl chloride) 0.085 55.241.50.06151.10.10061.30.130 45.30.071 1-Butanol + Poly(12% vinyl acetate-co-vinyl chloride) $t = 60 \ ^{\circ}\text{C}$ $t = 80 \ ^{\circ}\text{C}$ 2.70.007 12.70.009 54.80.06281.3 0.12891.50.1838.0 0.02212.30.040 15.20.046 0.021 0.092 0.231 5.20.016 0.033 26.468.5 84.8 0.14797.9 10.813.90.04440.70.0371,2-Dichloroethane + Polyacrylonitrile Acetone + Poly(styrene 45%-co-butadiene) $t = 70 \ ^{\circ}\text{C}$ $t = 50 \ ^{\circ}\text{C}$ 0.025 30.80.04340.70.05046.00.05816.30.088 0.031 13.70.004 45.30.07451.159.6 0.137 21.236.50.04643.90.05449.10.061 0.02246.5 0.078 0.092 59.70.130 26.70.040 26.552.336.3 0.046 47.20.07453.30.095 61.3 0.141 $t = 80 \ ^{\circ}\text{C}$ 0.061 41.547.30.07155.20.10522.40.017 36.3 0.02450.10.044 59.50.057 $t = 60 \ ^{\circ}\text{C}$ 27.30.02045.70.038 57.60.05360.70.060 12.70.009 54.80.069 81.3 0.14091.5 0.197 26.40.023 68.50.101 84.8 0.163 97.9 0.2581,2-Dichloroethane + Polystyrene 40.70.044 $t = 70 \ ^{\circ}\mathrm{C}$ Acetone + Poly(50% styrene-co-butyl methacrylate) 16.3 0.064 30.8 0.15240.70.244 46.0 0.306 21.20.092 36.50.209 43.90.289 49.10.352 $t = 60 \ ^{\circ}\mathrm{C}$ 26.70.1278.9 0.005 43.3 0.04481.7 0.133 95.9 0.201 16.8 0.013 0.053 98.7 0.224 $t = 80 \ ^{\circ}\text{C}$ 51.6 85.3 0.144 23.50.022 60.3 0.071 90.4 0.164 100.40.25422.40.070 36.3 0.12550.10.25759.50.360 27.30.221 0.027 67.6 0.090 0.090 45.757.6 0.33160.70.377 34.71,2-Dichloroethane + Poly(70% styrene-co-acrylonitrile) Acetonitrile + Polyacrylonitrile $t = 60 \ ^{\circ}\text{C}$ $t = 70 \ ^{\circ}\text{C}$ 30.8 6.7 0.011 18.30.03232.50.061 37.9 0.076 16.3 0.009 0.139 40.70.24346.00.296 21.20.026 36.523.20.196 0.28113.30.022 0.04143.949.10.339 26.70.048Acetonitrile + Poly(acrylonitrile-co-butadiene) $t = 80 \ ^{\circ}\text{C}$ 22.40.009 0.376 Acetonitrile + Poly(21% acrylonitrile-co-butadiene) 36.3 0.090 0.27959.550.127.30.039 45.70.25157.6 0.348 60.70.389 $t = 60 \,^{\circ}\text{C}$ Chloroform + Poly(33% acrylonitrile-co-butadiene) 6.8 0.011 26.10.065 34.10.11239.3 0.18713.70.02630.10.085 $t = 60 \ ^{\circ}\text{C}$ 13.6 0.080 0.1720.372 66.3 0.485 Acetonitrile + Poly(33% acrylonitrile-co-butadiene) 26.952.826.80.17039.6 0.26259.9 0.42473.3 0.563 $t = 60 \ ^{\circ}\mathrm{C}$ 6.70.023 18.3 0.065 37.9 0.206 32.50.148Chloroform + Poly(cis-1,4-butadiene) 13.30.043 23.20.091 $t = 60 \ ^{\circ}\text{C}$ Acetonitrile + Poly(51% acrylonitrile-co-butadiene) 13.6 0.114 26.9 0.228 52.80.44466.3 0.54826.80.224 39.6 0.494 0.615 0.33659.9 73.3 $t = 60 \ ^{\circ}\text{C}$ 6.8 0.021 26.10.111 34.10.182 39.30.293Chloroform + Poly(butyl methacrylate) 30.113.70.0470.144 $t = 70 \ ^{\circ}\text{C}$ Acetonitrile + Poly(cis-1, 4-butadiene)13.3 0.093 40.10.268 65.5 86.0 0.5660.43426.40.17653.60.34778.8 0.520 $t = 60 \ ^{\circ}\text{C}$ 0.010 18.30.017 32.50.030 37.90.039 6.723.213.30.010 0.024

Table 2. New Experimental Vapor-Liquid Equilibrium Data for Polymer + Solvent Systems (w_1 = Mass Fraction Solvent in Polymer Phase, t = Temperature, and P = Pressure)

P/kPa	w_1	P/kPa	w_1	P/kPa	w_1	P/kPa	w_1	P∕kPa	w_1	P/kPa	w_1	P/kPa	w_1	P/kPa	w_1		
	C	Chlorofo	rm + Pol	y(ethyle	ne oxide)		Cyclohexane + Poly(51% acrylonitrile-co-butadiene)									
			t = 6	0°C							t = f	so °C					
14.0	0.082	53.2	0.459	66.1	0.540	79.2	0.585	13.7	0.007	26.1	0.024	32.0	0.032	39.7	0.060		
28.1	0.276	58.4	0.497	72.3	0.567	87.1	0.609	19.1	0.011	26.4	0.024	35.6	0.047	43.1	0.091		
40.1	0.379							19.6	0.015								
Cł	loroform	+ Poly	50% styr	rene <i>-co-</i> l	outyl me	thacryla	te)		Су	clohexar	ne + Poly	v(cis-1,4-	butadie	ne)			
			+ 7	ഹംഗ							<i>t</i> = 6	2° 0					
13.3	0.078	40.1	0.222	65.5	0.369	86.0	0.512	13.5	0.082	26.9	0.189	33.6	0.272	38.1	0.338		
26.4	0.148	53.6	0.292	78.8	0.455			20.3	0.134				-				
	Chle	roform	$\pm Poly(e)$	turono-a	o-butadi	ane)			Cycl	hovene	$\pm Poly($	turene.	o-butad	iene)			
	Cinc	10101111	T OIY(S	lyrene-t	o-butaun	ene)			Cych	JIIEAAIIE	1 1 019(2	styrene-	o-butau	lene)			
	Chlore	oform +	Poly(5%	styrene	-co-buta	liene)			Cycloh	exane +	Poly(5%	b styrene	e-co-buta	(diene)			
			t = 5	0 °C							t = 6	50 °C					
8.5	0.040	31.6	0.272	47.6	0.489	53.7	0.580	4.8	0.027	19.9	0.120	29.5	0.189	37.6	0.276		
16.3	0.106	38.3	0.341	50.0	0.537	56.4	0.638	10.5	0.057	24.8	0.152	33.7	0.233	40.7	0.323		
29.9	0.204	40.0	0.457					10.9	0.103								
	Chloro	form +	Poly(23%	styren	e-co-buta	diene)			Cycloh	exane +	Poly(23	% styren	e-co-but	adiene)			
			t = 5	0 °C							t = 23	3.5 °C					
8.5	0.046	31.6	0.270	47.6	0.482	53.7	0.566	2.00	0.015	4.53	0.074	7.64	0.197	10.03	0.328		
16.3 25.5	$0.111 \\ 0.206$	38.3 45.5	0.338 0.449	50.0	0.514	20.4	0.621	2.99	0.035	0.83	0.141	9.03	0.276				
2010	0.200		•••••								$t = \epsilon$	50 °C					
	Chloro	form +	Poly(45%	b styren	e-co-buta	diene)		4.8	0.018	16.9	0.088	29.5	0.168	37.6	0.244		
			t = 5	0 °C				10.5	0.052	24.0	0.130	əə. i	0.208	40.7	0.207		
8.5	0.043	31.6	0.272	47.6	0.500	53.7	0.592		Cycloh	exane +	Poly(45	% styren	e-co-but	adiene)			
16.3 25 5	0.107	38.3	0.343	50.0	0.534	56.4	0.650				+ - 2	35 °C					
29.9	0.205	40.0	0.403					2.00	0.011	4.53	0.063	7.64	0.176	10.03	0.301		
		Chlorof	orm + Po	oly(vinyl	acetate)			2.99	0.028	6.83	0.127	9.03	0.253				
			t = 6	n °C							t = 6	80 °C					
14.0	0.088	53.2	0.334	66.1	0.416	79.2	0.466	4.8	0.021	19.9	0.103	29.5	0.163	37.6	0.240		
28.1	0.150	58.4	0.371	72.3	0.444	87.1	0.490	10.5	0.050	24.8	0.130	33.7	0.200	40.7	0.282		
40.1	0.245							16.9	0.087								
	Chlore	oform +	Poly(vin	yl aceta	te <i>-co-</i> eth	ylene)			Cycloh	exane +	Poly(vii	nyl aceta	te-co-eth	nylene)			
	Chlorof	orm + P	olv(9% v	invl ace	tate-co-et	thvlene)			Cyclobey	ane + P	olv(25%	vinvl ac	etate-co-	ethvlene	a)		
	emoron	51111 1 1	019(0707			ling renie)			$t = 80 ^{\circ}\mathrm{C}$								
14.0	0.021	53.0	t = 6	50 °C 66 1	0 196	79.2	0.935	75	0.011	30.7	t = 8	55 5	0.238	64 1	0 300		
28.1	0.067	58.4	0.147	72.3	0.216	87.1	0.253	14.4	0.037	54.9	0.228	60.7	0.200	70.4	0.369		
40.1	0.104							26.5	0.074								
	Chlorofo	rm + Pc	oly(25% v	vinyl ace	etate-co-e	thylene)		Cyclohex	ane + P	oly(50%	vinyl ac	etate-co-	ethylene	e)		
			+	50 °C				$t = 80 \ ^{\circ}\mathrm{C}$									
13.9	0.062	39.9	0.228	59.9	0.358	73.1	0.441	7.5	0.005	39.7	0.104	55.5	0.170	64.1	0.225		
26.7	0.140	53.2	0.320	66.5	0.401	79.3	0.481	14.4	0.025	54.9	0.159	60.7	0.205	70.4	0.287		
	Chlorofo	rm + Pc	1v(50% v	vinvl ace	tate-co-e	thylene)	26.5	0.046								
	emoroio			ing i dec		, un y reme	/		Cyclohex	ane + P	oly(70%	vinyl ac	etate-co-	ethylene	e)		
190	0.008	30.0	$t = \epsilon$	50 °C	0 439	73 1	0.519				+ = \$	20.00					
26.7	0.098	53.5 53.2	0.303	66.5	0.439 0.477	79.3	0.552	7.5	0.004	39.7	0.081	55.5	0.133	64.1	0.191		
	~						、 、	14.4	0.026	54.9	0.123	60.7	0.156	70.4	0.242		
	Chlorofo	rm + Po	oly(70% v	rinyl ace	etate-co-e	thylene)	26.5	0.042								
			$t = \epsilon$	50 °C						Hexa	ne + Po	lyacrylor	nitrile				
13.9	0.059	39.9	0.287	59.9	0.450	73.1	0.526				+ (20 °C					
40. <i>1</i>	0.101	00.2	0.403	00.0	0.409	13.0	0.075	12.4	0.001	25.1	0.007	36.0	0.007	43.3	0.009		
	Cycloh	exane +	Poly(acr	ylonitri	le-co-but	adiene)		_						·			
(Cyclohex	ane + P	oly(21% :	acryloni	trile-co-h	utadien	e)		Hex	ane + Pe	oiy(acryl	onitrile-	co-butad	iene)			
	,		·	so •C			-		Hexan	e + Poly	(21% act	rylonitri	le-co-but	adiene)			
13.7	0.030	32.0	$\iota = 0$ 0.114	35.7	0.176	39.7	0.216				t = 0	50 °C					
19.6	0.062	35.6	0.177	38.1	0.210	43.1	0.283	12.9	0.025	38.8	0.082	50.8	0.134	63.7	0.236		
26.4	0.084							25.6	0.044	44.5	0.104	57.5	U.170	64.9	U.251		

Table 2. (Continued)

788 Journal of Chemical and Engineering Data, Vol. 40, No. 4, 1995

P/kPa	w_1	P/kPa	w_1	P/kPa	w_1	P/kPa	w_1	P/kPa	w_1	P/kPa	w_1	P/kPa	w_1	P/kPa	w_1	
Hexane + Poly(33% acrylonitrile-co-butadiene)								Pentane + Poly{acrylonitrile-co-butadiene)								
			t =	60 °C					Pentar	he + Poly	y(21% a)	rylonitri	le-co-bu	tadiene)		
12.4	0.002	25.1	0.017	36.0	0.026	43.3	0.037				<i>+</i> –	60 °C				
	Hexan	e + Poly	(51% ac)	rylonitri	le-co-but	tadiene)		22.8	0.008	65.5	0.033	97.3	0.055	124.8	0.078	
		2	`					36.0	0.016	84.0	0.046	113.3	0.070	137.6	0.089	
12.9	0.018	38.8	t = 0.029	60 °C 50 8	0.041	63 7	0.062	46.8	0.021							
25.6	0.020	44.5	0.033	57.5	0.044	64.9	0.062		Pentar	ne + Poly	y(33% ac	rylonitri	le <i>-co-</i> bu	tadiene)		
		Havano	+ Polv	cie-1 4-bi	itadiana						<i>t</i> =	റെ ംവ				
		HEAdile	(101y(C13-1, 1 -01	atautene			13.3	0.009	64.7	0.027	89.7	0.036	115.7	0.050	
10.4	0.000	05.1	t =	60 °C	0.115	40.0	0.1.40	26.9	0.014	77.5	0.032	101.7	0.044	130.1	0.055	
12.4	0.036	25.1	0.078	36.0	0.115	43.3	0.146	53.5	0.025							
	Н	lexane +	Poly(st	yrene-co-	butadie	ne)			Pentar	ne + Poly	y(51% ao	rylonitri	le-co-bu	tadiene)		
	Hex	ane + P	oly(23%	styrene-	co-butad	liene)					t =	60 °C				
			<i>i</i>	70 %				22.8	0.010	65.5	0.019	97.3	0.022	124.8	0.027	
13.9	0.014	38.5	0.063	61.5	0.162	75.2	0.262	36.0 46.8	0.009	04.0	0.022	113.3	0.020	137.0	0.034	
27.3	0.043	48.8	0.110							D .				、 、		
	Hex	ane + Pe	olv(45%	stvrene-	co-butad	liene)				Pentane	e + Poly(<i>cis-</i> 1,4-bi	utadiene	e)		
						,		10.0		a . =	<i>t</i> =	60 °C				
13.9	0.016	38.5	t = 0.061	70 °C 61 5	0 155	75.2	0.249	13.3 26 9	0.012 0.024	$\begin{array}{c} 64.7 \\ 77.5 \end{array}$	0.061 0.082	89.7 101 7	0.095 0.117	115.7 130.1	0.134	
27.3	0.038	48.8	0.103	01.0	0.100	10.2	0.240	53.5	0.046	11.0	0.002	101.7	0.117	100.1	0.100	
		Metha	nol + Pc	oly(vinyl	acetate)				P	entane +	- Poly(st	yrene-co-	-butadie	ne)		
			<i>t</i> =	80 °C					Pent	tane + P	oly(23%	styrene-	co-butad	liene)		
13.6	0.021	39.9	0.041	65.7	0.062	79.1	0.080									
28.4	0.030	52.5	0.053					22.8	0.008	65.5	t = 0.033	97.3	0.055	124.8	0.078	
Ν	Iethanol	+ Poly(10% vin	yl acetat	e-co-ving	yl chlorid	le)	36.0	0.016	84.0	0.046	113.3	0.070	137.6	0.089	
			t =	80 °C				46.8	0.021							
13.6	0.029	39.9	0.032	65.7	0.037	79.1	0.038		Pent	tane + P	oly(45%	styrene-	co-butad	liene)		
28.4	0.030	52.5	0.037								<i>t</i> =	60 °C				
	Oct	ane + Po	oly(acryl	lonitrile-c	o-butad	iene)		22.8	0.010	65.5	0.019	97.3	0.022	124.8	0.027	
	Octon	a + Polv	(91% ac	rulonitril	e-co-but	adiona		46.8	0.013	84.0	0.022	113.3	0.026	137.6	0.034	
	Octain	e r i oly	(21 /0 at	ryiomern	e-co-but	autene)			T	oluene +	Poly(st	yrene-co-	butadie	ne)		
9.00	0.026	5.09	t =	60 °C	0 111	9.41	0.159		Tal	$ana \perp D$	al. (990)	atomana	oo hutad	lione)		
2.90 4.19	0.030 0.052	5.08 5.91	0.064 0.079	0.91	0.111	0.41	0.156		1010	tene $+ r$	01y(23%	styrene-	co-butat	liene)		
	Ostan	a Dala	(5107	1 ~ i + i 1	h	a diama)		0.00	0.090	2.04	t =	35 °C	0.027	= 49	0 0 0 0	
	Octan	e + Poly	(01% ac	ryionitrii	e-co-but	autene)		2.08	0.029	$\frac{3.04}{4.36}$	0.032	4.83	0.037	0.40	0.038	
-	.		<i>t</i> =	60 °C		0.41			- - 1					•		
2.85 4.19	0.007	5.08 5.91	0.013 0.015	6.91	0.021	8.41	0.026		Tol	uene + F	oly(5%	styrene-c	o-butad	iene)		
		_	-								<i>t</i> =	35 °C				
		Penta	ane + Pe	oiyacrylo	nitrile			$\begin{array}{c} 0.93 \\ 2.08 \end{array}$	0.043 0.099	3.04 4.36	0.184	4.83	0.416	5.48	0.513	
			t =	60 °C				2.00	0.000	4.00	0.010					
13.3	0.005	64.7	0.010	89.7 101.7	0.010	115.7	0.011									
20.9 53.5	0.007	11.0	0.010	101.7	0.010	100.1	0.011									

by varying the solvent, copolymer composition, and temperature.

Brief Summary of Theory for Correlation. To model VLE of copolymer solutions, we have used a perturbed hard-sphere chain (PHSC) equation of state (Song et al., 1993, 1994a-c). For ordinary VLE at low pressures, it is often possible to use the Flory-Huggins theory for a reasonable correlation of polymer + solvent VLE data, but we would like to use a model that can describe both upper and lower critical solution temperatures and VLE phase diagrams at high pressures and account for solvent compressibility. For binary polymer solutions, the PHSC

theory successfully reproduces all types of observed liquidliquid equilibrium phase diagrams, including upper or lower critical solution temperatures (UCST or LCST), or both, and closed partial-miscibility loops (Song et al., 1994c).

The PHSC equation of state is applicable to fluid mixtures containing any number of components over the entire range of fluid conditions. Since the PHSC equation of state uses molecular parameters that have physical significance, it provides a potentially useful method for predicting thermodynamic properties outside the range of available data. The PHSC equation of state for heteronuclear chains is

$$\frac{p}{\varrho k_{\rm B}T} = 1 + \varrho \sum_{i=1}^{m} \sum_{j=1}^{m} x_i x_j [\sum_{k=1}^{r_i} \sum_{l=1}^{r_j} b_{ij,kl} g_{ij,kl}] - \sum_{i=1}^{m} x_i \sum_{k=1}^{r_i-1} [g_{ii,k,k+1} - 1] - \frac{\varrho}{k_{\rm B}T} \sum_{i=1}^{m} \sum_{j=1}^{m} x_i x_j [\sum_{k=1}^{r_i} \sum_{l=1}^{r_j} a_{ij,kl}]$$
(2)

where p = pressure, T = absolute temperature, $\varrho = \text{number}$ density, m = number of components, x = mole fraction, $k_B = \text{Boltzmann constant}$, a = attractive energy parameter, b = second virial coefficient of hard spheres, r = numberof hard spheres per chain, and g = pair radial distributionfunction of hard spheres at contact. The subscript ij,kldenote the pair containing the kth segment of component i and the lth segment of component j. Parameters a and bare temperature-dependent as given by the Song-Mason (1989) method. For pure solvents and homopolymers, these are

$$a = (2/3)\pi\sigma^3 \epsilon F_a(k_{\rm B}T/\epsilon) \tag{3}$$

$$b = (2/3)\pi\sigma^3 F_b(k_{\rm B}T/\epsilon) \tag{4}$$

where $\sigma = \text{distance of separation at the minimum potential}$ energy, $\epsilon = \text{well depth of the pair potential, and } F_{\text{a}}$ and F_{b} are universal functions.

Parameters a and b for a pair of dissimilar segments A and B are given as

$$a_{\rm AB} = (2/3)\pi\sigma^3{}_{\rm AB}\epsilon_{\rm AB}F_a(k_{\rm B}T/\epsilon_{\rm AB}) \tag{5}$$

$$b_{\rm AB} = (2/3)\pi \sigma^3{}_{\rm AB} F_b (k_{\rm B} T/\epsilon_{\rm AB}) \tag{6}$$

where

$$\sigma_{\rm AB} = (1/2)(\sigma_{\rm A} + \sigma_{\rm B})(1 - \lambda_{\rm AB}) \tag{7}$$

$$\epsilon_{\rm AB} = (\epsilon_{\rm A}\epsilon_{\rm B})^{1/2}(1-\kappa_{\rm AB}) \tag{8}$$

Parameters σ_A and ϵ_A pertain to segment A; κ_{AB} and λ_{AB} are adjustable intersegmental parameters.

The pair correlation function is

$$g_{ij,kl} = \frac{1}{(1-\eta)} + \frac{2}{3} \frac{\xi_{ij,kl}}{(1-\eta)^2} + \frac{1}{2} \frac{\xi_{ij,kl}}{(1-\eta)^3}$$
(9)

where the packing fraction is

$$\eta = (\varrho/4) \sum_{i=1}^{m} x_i \sum_{k=1}^{r_i} b_{i,k}$$
(10)

and

$$\xi_{ij,kl} = \frac{\varrho}{4} \left[\frac{b_{i,k} b_{j,l}}{b_{ij,kl}} \right]^{1/3} \sum_{i=1}^{m} x_j \sum_{k=1}^{r_i} b_{i,k}^{2/3}$$
(11)

In summary, each pure solvent or homopolymer is characterized by three molecular constants: r, segment number per molecule; σ , segment size; and ϵ , nonbonded segment pair interaction energy. These molecular constants are obtained from readily available pure-component data for thermodynamic properties such as vapor pressures, densities, and compressibilities (Song et al., 1994b). Pure-component parameters for 77 solvents and for 22 polymers have been collected by Lambert (1995). For each unlike pair of segments, usually one and only rarely two



Figure 2. VLE for cyclohexane + poly(ethylene-co-vinyl acetate) solutions: w_1 , mass fraction cyclohexane; points, new experimental data; lines, calculation using PHSC equation of state parameters [cyclohexane + polyethylene, $\kappa = 0$, $\lambda = -0.0344$; cyclohexane + poly(vinyl acetate), $\kappa = 0$, $\lambda = -0.005$; poly(vinyl acetate) + polyethylene, $\kappa = 0$, $\lambda = -0.164$].



Figure 3. VLE for chloroform + polystyrene, poly(styrene-co-butyl methacrylate), and poly(butyl methacrylate) solutions: w_1 , mass fraction chloroform; points, experimental data; lines, calculation using PHSC equation of state parameters [chloroform + polystyrene, $\kappa = 0$, $\lambda = -0.025$; chloroform + poly(butyl methacrylate), $\kappa = 0$, $\lambda = -0.05$; polystyrene + poly(butyl methacrylate), $\kappa = 0$, $\lambda = -0.04$].

binary parameters (κ and λ), are needed for the mixture; these can be determined from experimental mixture properties.

Results and Discussion

The PHSC equation of state has been applied to some data sets from the Experimental Section. The experimental molecular weights of the polymers were used in the calculations; however, if these were not available, a 100 000 molecular weight has been assumed because molecular weight has a negligible effect on the VLE for high-molecular-weight polymer solutions. Pure-component molecular parameters $(r, \epsilon, \text{ and } \sigma)$ were obtained from pure-component properties (Lambert, 1995; Song et al., 1995).

Figure 2 shows pressure versus weight fraction solvent in the polymer phase for the system cyclohexane + poly-(ethylene-co-vinyl acetate) at 80 °C. At a given pressure, solvent absorption in the copolymer increases as its ethylene content rises. This rise is expected because the hydrocarbon segments of cyclohexane are better liked by the similar hydrocarbon segments of ethylene, whereas polar segments of vinyl acetate are disliked by the solvent molecules. The solubility of cyclohexane falls in proportion to the vinyl acetate concentration in the copolymer. All binary parameters κ were set equal to zero, whereas three λ binary parameters were regressed using the experimental data. The PHSC equation of state can accurately correlate the experimental data.

Figure 3 shows VLE for a system containing chloroform and the copolymer poly(styrene-co-butyl methacrylate) and chloroform with the parent homopolymers polystyrene and



Figure 4. VLE for cyclohexane + poly(*cis*-1,4-butadiene) and poly-(butadiene-*co*-acrylonitrile) solutions: w_1 , mass fraction cyclohexane; points, new experimental data; lines, calculation using PHSC equation of state parameters [cyclohexane + polyacrylonitrile, κ = 0.082, λ = -0.08; cyclohexane + poly(*cis*-1,4-butadiene), κ = 0, λ = -0.021; poly(*cis*-1,4-butadiene) + polyacrylonitrile, κ = 0.027, λ = 0.0].



Figure 5. VLE for hexane + poly(*cis*-1,4-butadiene) poly(butadiene-co-acrylonitrile) solutions: w_1 , mass fraction hexane; points, experimental data; lines, calculation using PHSC equation of state parameters [hexane + polyacrylonitrile, $\kappa = 0.168$, $\lambda = -0.33$; hexane + poly(*cis*-1,4-butadiene), $\kappa = -0.023$; $\lambda = -0.02$; poly(*cis*-1,4-butadiene) + polyacrylonitrile, $\kappa = 0.027$, $\lambda = 0.0$].

poly(butyl methacrylate). Figure 3 shows relative pressure $P/P_{\rm sat}$ ($P_{\rm sat}$ = saturation pressure of the solvent at the given temperature) instead of P because data at 50 °C are compared with data at 70 °C; normalizing pressure with $P_{\rm sat}$ tends to eliminate the temperature effect in a small temperature range such as here. Going from polystyrene to poly(50% styrene-co-butyl methacrylate) to poly(butyl methacrylate) increases the solubility of chloroform because of favorable polar interaction between chloroform and butyl methacrylate segments. Chloroform also has some polar interaction with styrene segments, but these are not as strong as those with the butyl methacrylate segments. The solubility of chloroform rises in proportion to the butyl methacrylate content in the polymer.

For the system shown in Figure 3, all binary parameters κ were set equal to zero. Solvent + homopolymer λ parameters were obtained from their VLE experimental data. The polystyrene + poly(butyl methacrylate) λ parameter was obtained from the copolymer + solvent VLE data. With only one adjustable parameter for each data set, the PHSC equation of state can correlate the VLE behavior.

Figures 4–6 show VLE data for hydrocarbons with poly-(butadiene-co-acrylonitrile) at 60 °C. For these systems, solubility is not a simple function of copolymer composition.

Figure 4 compares VLE data for cyclohexane + poly-(butadiene-co-acrylonitrile) with theory. At a given pressure, progressing from polybutadiene to poly(butadieneco-12% acrylonitrile) to poly(butadiene-co-51\% acrylonitrile), the solubility of cyclohexane decreases dramatically with the acrylonitrile content in the copolymer. Cyclohexane



Figure 6. VLE for pentane + poly(butadiene-co-acrylonitrile) solutions: w_1 , mass fraction pentane; points, new experimental data; lines, calculation using PHSC equation of state parameters: [pentane + polyacrylonitrile, $\kappa = 0$, $\lambda = -0.06$; pentane + poly(cis-1,4-butadiene), $\kappa = 0$, $\lambda = -0.085$; poly(cis-1,4-butadiene) + polyacrylonitrile, $\kappa = 0.027$, $\lambda = 0$].

has negligible solubility in polyacrylonitrile due to strong repulsion between nonpolar cyclohexane segments and highly polar acrylonitrile segments. Binary parameters for the PHSC equation of state were regressed from the VLE data for cyclohexane in polybutadiene and poly(butadieneco-21% acrylonitrile) as shown in Figure 4. VLE for cyclohexane in poly(butadiene-co-51% acrylonitrile) were predicted; the prediction is in excellent agreement with the experimental data.

Figure 5 shows results for hexane + poly(butadiene-coacrylonitrile). In this case, VLE behavior is similar to that for cyclohexane + poly(butadiene-co-acrylonitrile) shown in Figure 4. The solubility of hexane decreases dramatically with the acrylonitrile content in the copolymer. PHSC can successfully correlate this behavior with adjustable binary parameters given in Figure 5. The predicted VLE for hexane + poly(butadiene-co-51% acrylonitrile) is in good agreement with experiment.

Finally, Figure 6 shows results for pentane with poly-(butadiene-co-acrylonitrile). Results are similar to those for hexane (Figure 5) and cyclohexane (Figure 4). The solubility of pentane decreases dramatically with acrylonitrile content in the copolymer. The PHSC equation of state can successfully correlate the data for pentane with only one binary parameter for each pair as shown in Figure 6. Polybutadiene + polyacrylonitrile binary interaction parameters were the same as those used in Figures 4 and 5.

Conclusions

New VLE data have been obtained for 43 copolymer + solvent binaries and for 14 homopolymer + solvent binaries, using a gravimetric absorption technique. "Expected" nonideal behavior has been obtained for systems such as cyclohexane + poly(ethylene-co-vinyl acetate) and chloroform + poly(styrene-co-butyl methacrylate). However, strong nonlinear nonideal behavior has been observed for systems containing hydrocarbons and poly(butadiene-coacrylonitrile).

The perturbed hard-sphere chain (PHSC) equation of state for mixtures of heteronuclear chains can represent vapor-liquid equilibria of real copolymer solutions using one or at most two binary parameters for each pair of unlike segments.

Acknowledgment

For experimental assistance, the authors are grateful to Gary Yeung, Bin W. Zhao, and Kathy Yi. Helpful discussions with David Soane, Toshiaki Hino, Yuhua Song, and Stephen Lambert are much appreciated.

Literature Cited

- Baker, R. W.; Yoshioka, N.; Mohr, J. M.; Kahn, A. J. Separation of Organic Vapors from Air. J. Membr. Sci. 1987, 31, 259. Ballantine, D. S., Jr.; Wohltjen, H. Surface Acoustic Wave Device for
- Chemical Analysis. Anal. Chem. 1989, 61, 704A-715A. Bawn, C. E. H.; Wajid, M. A. High Polymer Solutions. Part 7. Vapour Pressure of Polystyrene Solutions in Acetone, Chloroform and Propyl Acetate. Trans. Faraday Soc. 1956, 52, 1658.
- Bawn, C. E. H.; Freeman, R.; Kamaliddin, A. High Polymer Solutions. Part 1. Vapour Pressure of Polystyrene Solutions. Trans. Faraday Soc. 1950, 46, 677. Bonner, D. C.; Prausnitz, J. M. Thermodynamic Properties of Some

Concentrated Polymer Solutions. J. Polym. Sci. 1974, 21, 51-73. Corneliussen, R.; Rice, S. A.; Yamakawa, H. On the Thermodynamic

- Properties of Solutions of Polar Polymers. A Comparision of Experi-
- The and Theory J. Chem. Phys. 1963, 38, 1768-1778.
 Danner, R. P.; High, M. S. Handbook of Polymer Solution Thermody-namics; DIPPR, AIChE: New York, 1993.
- Grate, J. W.; Klusty, M.; McGill, R. A.; Abraham, M. H.; Whiting, G.; Andonian-Haftvan, J. The Predominant Role of Swelling-induced Modulus Changes of the Sorbent Phase in Determining the Responses of Polymer-Coated Surface Acoustic Wave Vapor Sensor. Anal. Chem. 1992, 64, 610-624. High, M. S.; Danner, R. P. Prediction of Solvent Activities in Polymer
- Solutions. Fluid Phase Equilib. 1990, 55, 1-15. Iwai, Y.; Arai, Y. Measurement and Prediction of Solubilities of
- Hydrocarbon Vapors in Molten Polymers. J. Chem. Eng. Jpn. 1989, 22, 155-161.
- Iwai, Y.; Miyamoto, S.; Ikeda, H.; Arai, Y. Measurement and Prediction of solubilities of Ethylbenzene vapor in Styrene-butadiene Rubbers. Fluid Phase Equilib. 1991, 68, 197-205. Kubota, H.; Yoshino, N.; Ogiwara, Y. Vapor Phase Photografting on
- Low-Density Film In Binary Monomer System. J. Appl. Polym. Sci. 1990, 39, 1231.
- Lambert, S. M. Molecular Thermodynamics of Liquid-Liquid Equilibria in Polymer Mixtures. Ph.D. Thesis, University of California, Berkeley, 1995.
- Matsumoto, K.; Ishii, K.; Kuroda, T.; Inoue, K.; Iwama, A. Membrane Process for Organic Vapor Recovery from Air. Polym. J. 1991, 23, 491-499.
- Maeda, Y.; Tsuyumoto, M.; Karakane, H.; Tsugaya, H. Separation of Water-Ethanol Mixture by Pervaporation Through Hydrolyzed Polyacrylonitrile Hollow Fibre Membrane. Polym. J. 1991, 23, 501-511
- Moore, W. R.; Shuttleworth, R. Thermodynamic Properties of Solutions of Cellulose Acetate and Cellulose Nitrate. J. Polym. Sci., Part A-1 1963, 733-749.

- Napper, D. H. Polymeric Stabilization of Colloidal Dispersions; Academic Press, Inc.: London, 1983. Panayiotou, C. P.; Vera, J. H. Thermodynamic of Polymer-Polymer-
- Solvent and Block Copolymer-Solvent Systems. I. Experimental Systems. Polym. J. 1984, 16, 89.
- Saeki, S.; Holste, J. C.; Bonner, D. C. Vapor-Liquid Equilibria for Polybutadiene and Polyisoprene Solutions. J. Polym. Sci., Polym. Ed. 1982, 20, 793
- Song, Y.; Mason, E. A. Statistical-Mechanical Theory of a New Analytical Equation of State. J. Chem. Phys. 1989, 91, 7840-7853.
- Song, Y., Lambert, S. M.; Prausnitz, J. M. Fluid-Phase Equilibria for olymer-Solvent Systems from a Perturbed Hard-Sphere-Chain Equation of State Based on the Integral Theory of Fluids. Annual AIChE Meeting, St. Louis, MO, Nov 1993. Song, Y., Lambert, S. M.; Prausnitz, J. M. Equation of State for
- Mixtures of Hard-Sphere Chain Including Copolymers. Macromolecules 1994a, 27, 441-448.
- Song, Y., Lambert, S. M.; Prausnitz, J. M. A Perturbed Hard-Sphere-Chain Equation of State for Normal Fluids and Polymers. Ind. Eng. Chem. Res. 1994b, 33, 1047-1057.
- Song, Y., Lambert, S. M.; Prausnitz, J. M. Liquid-Liquid Phase Diagrams for Binary Polymer Solutions from a Perturbed Hard-Sphere-Chain Equation of State. Chem. Eng. Sci. 1994c, 49, 2765-2775
- Song, Y.; Hino, T.; Lambert, S. M.; Prausnitz, J. M. Liquid-liquid equilibria for polymer solutions and blends, including copolymers. Submitted for publication to Fluid Phase Equilib. 1995.
- Wang, K.; Cheng, Y.; Fu, J.; Hu, Y. Vapor-Liquid Equilibria for Mixtures of Toluene and Polystyrene with Different Molecular Weights. Chin. J. Chem. Eng. 1993, 1, 65-73.
- Wohlfarth, C. Calculation of phase equilibria in random copolymer systems. Makromol. Chem. Theory Simul. 1993, 2, 605-635.
- Wohlfarth, C. Vapour-liquid equilibrium data of binary polymer solutions, vapour-pressures, henry-constants and segment-molar excess gibbs free energies; Elsevier: Amsterdam, 1994.

Received for review February 6, 1995. Accepted March 7, 1995.[®] This work was supported by the Director, Office of Energy Research, Office of Basic Energy Science, Chemical Sciences Division of the U.S. Department of Energy under Contract No. DE-AC03-76SF0098. For additional financial support, the authors are grateful to the donors of the Petroleum Research Fund, administered by the American Chemical Society, E.I. du Pont de Nemours (Philadelphia, PA), and Koninklijke Shell (Amsterdam, The Netherlands).

JE950036K

[®] Abstract published in Advance ACS Abstracts, June 1, 1995.